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Synthesis and magnetic properties of 12L-perovskites $Ba_4LnIr_3O_{12}$ (Ln = lanthanides)

Yuki Shimoda, Yoshihiro Doi, Makoto Wakeshima, Yukio Hinatsu*

Division of Chemistry, Hokkaido University, Sapporo 060-0810, Japan

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ABSTRACT

New quadruple perovskite oxides $Ba_4LnIr_3O_{12}$ (Ln= lanthanides) were prepared and their magnetic properties were investigated. They crystallize in the monoclinic 12L-perovskite-type structure with space group C2/m. The Ir_3O_{12} trimers and LnO_6 octahedra are alternately linked by corner-sharing and form the perovskite-type structure with 12 layers. The Ln and Ir ions are both in the tetravalent state for Ln= Ce, Pr, and Tb compounds ($Ba_4Ln^{4+}Ir_3^{4+}O_{12}$), and for other compounds (Ln= La, Nd, Sm-Gd, Dy-Lu), Ln ions are in the trivalent state and the mean oxidation state of Ir ions is $+4.33(Ba_4Ln^{3+}Ir_3^{4+.33}O_{12})$. An antiferromagnetic transition has been observed for Ln= Ce, Pr, and Tb compounds at 10.5, 35, and 16 K, respectively, while the other compounds are paramagnetic down to 1.8 K.

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1. Introduction

The perovskite oxides ABO_3 form a wide family of compounds, reflecting the flexibility in the chemical composition and crystal structure. Structures of perovskite compounds can be regarded as the stacking of close-packed AO_3 layers and the filling of subsequent octahedral sites by B-site ions. The difference in the stacking sequence changes the way of linkage of BO_6 octahedra: the corner-sharing BO_6 in the cubic perovskite (3L: three-layer) with abc... sequence, the face-sharing BO_6 in 2L-perovskite (2L: two-layer) with ab... sequence, and mixed linkages between the corner- and face-sharing in various intergrowth structures [1].

The 6L-perovskites $Ba_3MM'_2O_9$ (M= alkali metals, alkaline earth elements, 3d transition metals, lanthanides; M'=Ru, Ir) have been investigated [2–29]. In this structure, two $M'O_6$ octahedra are connected by face-sharing and form an M'_2O_9 dimer. These dimers and MO_6 octahedra are placed alternately; thus, 6-layer (6L) structure is generated. The stacking sequence of AO_3 layers is abacbc... For many of these compounds, an antiferromagnetic spin-pairing occurs in the M'_2O_9 dimer even at room temperature. In addition, the $Ba_3MM'_2O_9$ compounds show magnetic transitions at low temperatures, which originates from the magnetic interaction between M and M' ions.

Then, we focused our attention on new quadruple perovskite compounds $Ba_4LnRu_3O_{12}$ (Ln = lanthanides) in which the ratio of Ln:Ru is 1:3. In the Ba₄LnRu₃O₁₂, three RuO₆ octahedra are faceshared, forming a Ru₃O₁₂ trimer, and we thought that peculiar magnetic behavior due to new alignment of the *Ln* and Ru ions should be observed [30]. In the Ba₃LnRu₂O₉, the ground state of the total spin of the isolated Ru₂O₉ dimer may be zero, i.e., $S_{total} = S_1 + S_2 = 0$, for the case that the antiferromagnetic coupling exists between the Ru ions. On the other hand, in the case of Ba₄LnRu₃O₁₂ compounds, the total magnetic moment of the Ru₃O₁₂ trimer does not disappear as far as three Ru ions are equivalent. The results of the magnetic susceptibility and specific heat measurements are that although no long-range magnetic ordering of Ru⁴⁺ ions was found for the 6L-perovskite Ba₃TbRu₂O₉, Ba₄TbRu₃O₁₂ showed an antiferromagnetic transition at 24 K, which is due to the relatively strong magnetic interaction between Tb⁴⁺ ions and Ru₃O₁₂ trimer via the Tb-O-Ru pathway owing to an uncancelled magnetic moment in the Ru₃O₁₂ trimer.

In this work, we challenged preparing new iridium-containing quadruple perovskite compounds $Ba_4LnIr_3O_{12}$. Battle et al. reported that the compounds $Ba_4MIr_3O_{12}$ (M=Li, Na, Mg, Y, Lu, Zr, and Ce) adopted the 6L-perovskite structure, i.e., the face-sharing octahedral were occupied solely by Ir atoms, and the corner-sharing octahedral were occupied statistically by the remaining Ir and the M atoms in a 1:3 ratio [31]. Although these M atoms have different charges and sizes, all the $Ba_4MIr_3O_{12}$ compounds had the same crystal structure. We could introduce the Ir atoms in a 3:1 ratio with lanthanides into the B-site of the

^{*} Corresponding author. Fax: +81117062702. E-mail address: hinatsu@sci.hokudai.ac.jp (Y. Hinatsu).

perovskite, and stabilize the fully ordered 12L perovskite structure. In order to elucidate the properties of Ir_3O_{12} trimer, the magnetic properties of these $Ba_4LnIr_3O_{12}$ compounds were investigated.

2. Experimental

2.1. Synthesis

Polycrystalline samples of $Ba_4LnIr_3O_{12}$ (Ln = La-Nd, Sm-Lu) were prepared by the standard solid-state reaction. BaIrO₃, $BaLnO_3$ (Ln = Ce, Pr, Tb), BaO, BaO_2 , and Ln_2O_3 (Ln = La, Pr, Sm-Gd, Dy-Lu) were used as starting materials. Among them, La₂O₃ and Nd₂O₃ absorb moistures in air and easily form lanthanide hydroxides *Ln*(OH)₃. Then, these oxides were heated at 900 °C for 12 h so as to decompose into Ln_2O_3 before use. It is known that both the ternary oxides BaIrO₃ and BaLnO₃ are formed with stoichiometric compositions. BalrO₃ was prepared by the solid state reaction of BaCO₃ and IrO₂ at 900 °C for 12 h and at 1100 °C for 12 h [32,33]. Ba $LnO_3 (Ln = Ce, Pr, Tb)$ were obtained by heating mixtures of BaCO₃ and CeO₂ (or Pr₆O₁₁, Tb₄O₇) in an oxygen flowing atmosphere [34-36]. These starting materials were weighed out in the appropriate metal ratio, and well mixed in an agate mortar. The mixtures were pressed into pellets and enclosed with platinum tubes, and they were sealed in evacuated silica tubes. Then, they were fired at 1250 °C for 36 h.

The obtained phases were identified by powder X-ray diffraction (XRD) measurements. Impurity phases such as 6L-perovskite $Ba_3LnIr_2O_9$ [25,26] were formed due to the sublimation of Ba oxides and Ir oxides. In order to decrease such impurity phases, the excess amount ($\sim\!30\%$) of $BaIrO_3$ [32,33] was added during sample preparation.

2.2. X-ray diffraction analysis

Powder X-ray diffraction profiles were measured using a Rigaku Multi-Flex diffractometer with $\text{Cu}K\alpha$ radiation equipped with a curved graphite monochromator. The data were collected by step-scanning in the angle range of $10^{\circ} \le 2\theta \le 120^{\circ}$ at a 2θ step-size of 0.02° . The X-ray diffraction data were analyzed by the Rietveld technique, using the programs RIETAN2000 [37].

2.3. Magnetic susceptibility measurements

The temperature -dependence of the magnetic susceptibility was measured in an applied field of 0.1 T over the temperature range of $1.8~\rm K \le T \le 400~\rm K$, using a SQUID magnetometer (Quantum Design, MPMS5S). The susceptibility measurements were performed under both zero-field-cooled (ZFC) and field-cooled (FC) conditions. The former was measured upon heating the sample to $400~\rm K$ under the applied magnetic field of $0.1~\rm T$ after zero-field cooling to $1.8~\rm K$. The latter was measured upon cooling the sample from $400~\rm to~1.8~\rm K$ at $0.1~\rm T$.

2.4. Specific heat measurements

Specific heat measurements were performed using a relaxation technique by a commercial heat capacity measuring system (Quantum Design, PPMS) in the temperature range of 1.8–400 K. The sintered sample in the form of a pellet was mounted on a thin alumina plate with Apiezon grease for better thermal contact.

3. Results and discussion

3.1. Synthesis and analysis of the structure

We succeeded in synthesizing new perovskite-related compounds $Ba_4LnIr_3O_{12}$ (Ln=La-Nd, Sm-Lu). Fig. 1 shows the powder X-ray diffraction profile of $Ba_4NdIr_3O_{12}$ as an example. To analyze the XRD diffraction profile, we used the crystal structure data of $Ba_4LnRu_3O_{12}$ as the initial structural model. The $Ba_4LnRu_3O_{12}$ for Ln=Tb-Lu have a hexagonal unit cell with space group R-3m (No. 166), while those for Ln=La-Nd, Sm-Gd have a monoclinic unit cell with space group C2/m (No. 12) [30]. In the case that the space group R-3m were applied, the X-ray diffraction lines observed at $2\theta=27^\circ$ and 30° , for example, should not be split. However, experimental results show that each of these diffraction lines is split into two diffraction lines, as shown in the inset of Fig. 1. All the $Ba_4LnIr_3O_{12}$ compounds prepared in this study were indexed with a monoclinic 12L-perovskite type cell with space group C2/m.

Fig. 2 shows the crystal structure of $Ba_4LnIr_3O_{12}$. Three IrO_6 octahedra are connected to each other by face-sharing and form an Ir_3O_{12} trimer. The Ir_3O_{12} trimers and LnO_6 octahedra are alternately linked by corner-sharing. As shown in this figure, perovskite-type structure with 12 layers is formed (the stacking sequence: ababcacabcbc...). Similar structures have been reported for $Ba_4ZrRu_3O_{12}$ [38] and $Ba_4LnMn_3O_{12}$ (Ln = Ce, Pr) [39]. The distances between Ir atoms in the Ir_3O_{12} trimer are determined to be 2.48-2.60 Å, which are much shorter than double the metallic radius of Ir (2.72 Å) [40]. This means that a strong interaction between Ir ions should exist in the trimer. The oxidation state of Ln and Ln are both tetravalent for Ln = Ce, Pr, Tb compounds $(Ba_4Ln^{4+}Ir_3^{4+}O_{12})$, and Ln ions are in the trivalent state and the mean oxidation state of Ln ions is +4.33 for other compounds $(Ba_4Ln^{3+}Ir_3^{4+33}O_{12})$.

Fig. 3 shows the variation of lattice parameters with the ionic radius of Ln^{3+} . Except for the compounds having the Ln^{4+} ion (Ln= Ce, Pr, and Tb), the lattice parameters a, b, and c monotonously increase with the Ln^{3+} ionic radius. The angle β increases from 90° with the Ln^{3+} ionic radius, which indicates that with increasing the size of the Ln^{3+} ion, the crystal structures of $Ba_4LnIr_3O_{12}$ are more distorted because of the larger difference in the ionic radius between Ln and Ir ions.

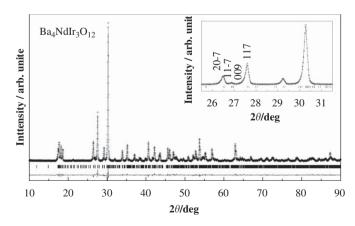


Fig. 1. X-ray diffraction profiles of Ba₄Ndlr₃O₁₂. The calculated and observed profiles are shown on the top solid line and cross markers, respectively. The vertical marks in the middle show positions calculated for Bragg reflections. The lower trace is a plot of the difference between calculated and observed intensities. The inset shows the profiles between $25^{\circ} \le 20 \le 32^{\circ}$. The diffraction lines are indexed with a monoclinic 12L-perovskite type cell.

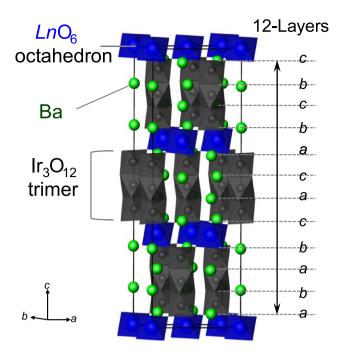


Fig. 2. The schematic crystal structure of Ba₄LnIr₃O₁₂.

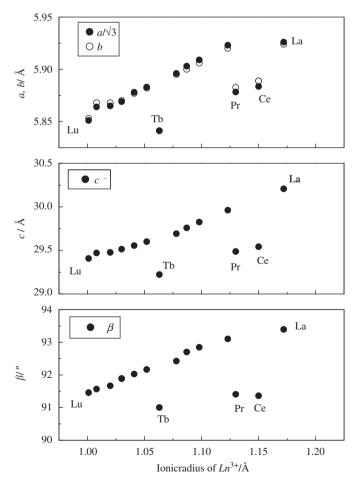


Fig. 3. The variation of lattice parameters for Ba₄LnIr₃O₁₂ against the ionic radius of Ln^{3+} .

3.2. Magnetic properties

3.2.1. $(Ba_4Ln^{3+}Ir_3^{4+.33}O_{12})(Ln=La,Nd-Gd,Dy-Lu)$ Figs. 4(a) and (b) show the temperature dependence of the reciprocal magnetic susceptibility for Ba₄LnIr₃O₁₂ (Ln = La, Nd-Gd, Dy-Lu). These compounds are paramagnetic down to 1.8 K. In the temperature range where the Curie-Weiss law holds, the effective magnetic moments were determined and they are listed in Table 1. The effective magnetic moments determined for $Ba_4LnIr_3O_{12}$ (μ_{eff}) are close to the magnetic moments of Ln^{3+} ions (μ_{Ln}) , i.e., the contribution of the trimer $Ir_3^{4+.33}O_{12}$ to the magnetic properties of Ba₄LnIr₃O₁₂ is negligible. This result can be understood in the following.

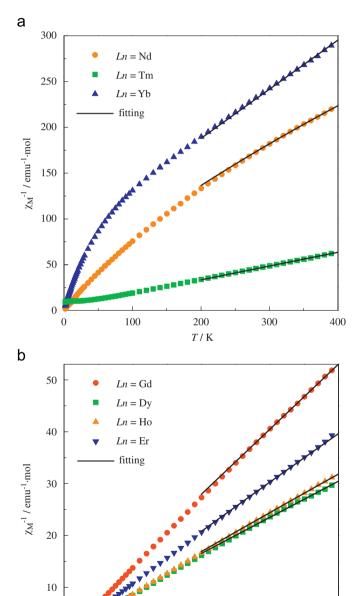


Fig. 4. (a) Temperature dependence of the reciprocal magnetic susceptibilities for $Ba_4LnIr_3O_{12}$ (Ln = Nd, Tm, Yb). The solid line is the Curie-Weiss fitting. (b) Temperature dependence of the magnetic susceptibilities for $Ba_4LnIr_3O_{12}$ (Ln = Gd, Dy-Er). The solid line is the Curie-Weiss fitting.

200

T/K

300

400

100

0

Table 1Magnetic properties for Ba₄*Ln*Ir₃O₁₂.

Ln ³⁺	Electronic configuration	J	$\mu_{ m eff}\left(\mu_{ m B} ight)$	$\mu_{Ln}^{a}(\mu_{B})$	Magnetic properties
La ³⁺	4f ⁰	0	_	_	Diamagnetic
Nd ³⁺	$4f^3$	9/2	3.60	3.62	Curie-Weiss
Sm ³⁺	4f ⁵ 4f ⁶	5/2		0.84	van Vleck
Eu ³⁺	$4f^6$	0	_	_	van Vleck
Gd ³⁺	$4f^7$	7/2	7.98	7.94	Curie-Weiss
Dy ³⁺	$4f^9$	15/2	10.71	10.63	Curie-Weiss
Ho ³⁺	$4f^{10}$	8	10.36	10.60	Curie-Weiss
Er ³⁺	$4f^{11}$	15/2	9.21	9.59	Curie-Weiss
Tm ³⁺	$4f^{12}$	6	7.31	7.57	Curie-Weiss
Yb ³⁺	$4f^{13}$	7/2	3.86	4.54	Curie-Weiss
Lu ³⁺	$4f^{14}$	0	_	_	Diamagnetic
Ln ⁴⁺	Electronic configuration	J	$\mu_{ m eff}\left(\mu_{ m B} ight)$	μ _{cal} ^b (μ _B)	Magnetic properties (T_N)
Ce ⁴⁺	4f ⁰	0	1.61	1.73	Antiferromagnetic (10.5 K)
Pr ⁴⁺	$4f^1$	5/2	2.94	3.07	Antiferromagnetic (35 K)
Tb ⁴⁺	$4f^7$	7/2	8.02	8.13	Antiferromagnetic (16 K)

^a μ_{In} : free ion values for Ln^{3+} .

The distances between Ir atoms in the Ir₃O₁₂ trimer are 2.48- $2.60\,\text{Å}$. The short Ir–Ir interatomic distances in the Ir₃O₁₂ trimer suggest the overlap of metal d orbitals having lobes along the threefold symmetry axis, which means the formation of molecular orbitals in the Ir₃O₁₂ trimer. The electronic structure of Ru₃Cl₁₂ with D_{3d} point symmetry has been described [41]. The energy scheme of Ir₃O₁₂ in the Ba₄LnIr₃O₁₂ should be similar to the case of Ru₃Cl₁₂, but the degenerated energy levels are expected to be split into more levels due to the monoclinic distortion of the Ir₃O₁₂ trimer and to the spin-orbit coupling of the 5d electrons (Ir ions). The electronic configuration of the $Ir_3^{4+.33}O_{12}$ trimer (the number of 5d electrons is 14) with D_{3d} point symmetry is $(a_{1g})^2(e_g)^4(a_{2u})^2(e_u)^4(e_g)^2$. Fig. 5 illustrates the schematic energy level diagrams for the Ir–Ir interactions in (a) $Ir_{3}^{4+.33}O_{12}$ and (b) $Ir_3^{4+}O_{12}$. The highest occupied e_g orbital of the Ir_3O_{12} trimer should be split into two singlets by the monoclinic distortion, which causes the S = 0 ground state of the filled HOMO level. Therefore, the Ir₃^{4+.33}O₁₂ trimer does not contribute to the magnetic properties of $Ba_4LnIr_3O_{12}$ (Ln = La, Nd-Gd, Dy-Lu), and both Ba₄LaIr₃O₁₂ and Ba₄LuIr₃O₁₂ are diamagnetic.

3.2.2. $Ba_4Ln^{4+}Ir_3^{4+}O_{12}$. (Ln = Ce, Pr, Tb)

The temperature dependence of the magnetic susceptibility and the reciprocal susceptibility for Ba₄CeIr₃O₁₂ is shown in Figs. 6(a) and (b), respectively. The same temperature dependence of the susceptibility and the reciprocal susceptibility for Ba₄PrIr₃O₁₂ and Ba₄TbIr₃O₁₂ is depicted in Figs. 7 and 9, respectively. From the Curie–Weiss fitting in the temperature range between 300 and 400 K, the effective magnetic moments of Ba₄LnIr₃O₁₂ with Ln = Ce, Pr, and Tb are obtained to be 1.24, 2.94, and 8.02 $\mu_{\rm B}$, respectively (Table 1). When the magnetic ions in these compounds are magnetically independent, the effective magnetic moments of Ba₄LnIr₃O₁₂ should be calculated from the equation $\mu_{cal} = \sqrt{\mu_{Ln^{4+}}^2 + 3 \times \mu_{lr^{4+}}^2}$. However, the effective magnetic moments obtained experimentally are far below the calculated moments. We will discuss the magnetic behavior of Ir₄⁴+O₁₂ trimer in the same way as the case of Ir₃^{4+.33}O₁₂. The Ir₄⁴⁺O₁₂ trimer has fifteen 5d electrons, and its electronic

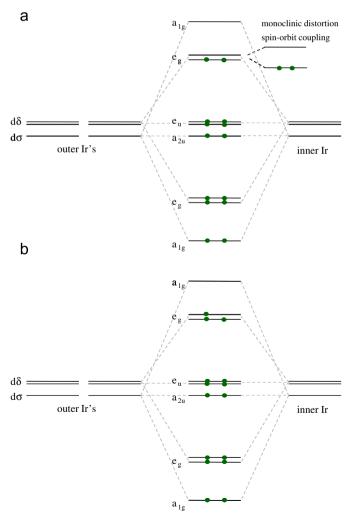
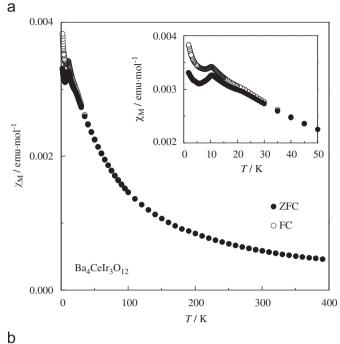


Fig. 5. Schematic energy level diagrams for (a) $Ir_3^{4+,33}O_{12}$ trimer and (b) $Ir_3^{4+}O_{12}$ trimer

^b μ_{cal} : calculated moments for Ln^{4+} and S=1/2 (see text).



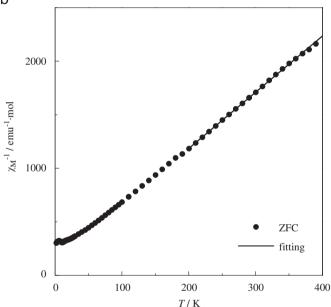
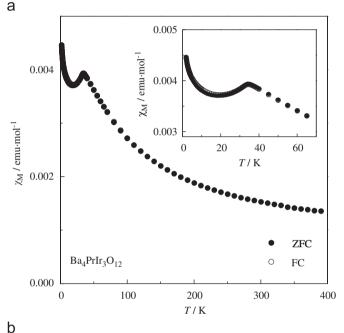


Fig. 6. (a) Temperature dependence of the magnetic susceptibilities for $Ba_4Celr_3O_{12}$. The inset shows the magnetic susceptibilities at low temperatures. (b) The reciprocal magnetic susceptibility against temperature. The solid line is the Curie–Weiss fitting.

configuration is $(a_{1g})^2(e_g)^4(a_{2u})^2(e_u)^4(e_g)^3$, which should cause the S=1/2 ground state (see Fig. 5(b)). In this case, the effective magnetic moments for (Ln=Ce,Pr,Tb) are calculated from the equation $\mu_{cal}=\sqrt{\mu_{Ln^{4+}}^2+\mu_{S=1/2}^2}$, and they are listed in Table 1. The effective magnetic moments of $\text{Ba}_4Ln\text{Ir}_3\text{O}_{12}$ experimentally determined are almost in accordance with the calculated moments. The moment of $\text{Ba}_4\text{Celr}_3\text{O}_{12}$ is a little smaller than the expected value, which is due to existence of impurities such as $\text{Ba}_3\text{Celr}_2\text{O}_9$. The $\text{Ba}_3\text{Celr}_2\text{O}_9$ shows a very small effective magnetic moment, because the magnetic moments of Ir ions cancel each other in the Ir_2O_9 dimer [25]. In the following, magnetic properties of each $\text{Ba}_4Ln\text{Ir}_3\text{O}_{12}$ (Ln=Ce,Pr,Tb) compound at lower temperatures will be described.

 $3.2.2.1.\ Ba_4 Celr_3 O_{12}$ Magnetic susceptibility measurements on Ba₄Celr₃O₁₂ show that an antiferromagnetic transition occurs at 10.5 K (see Fig. 6(a)). On the other hand, ruthenium-containing compound Ba₄CeRu₃O₁₂ is paramagnetic down to 1.8 K [30]. Since the tetravalent cerium ion is diamagnetic, the S=1/2 state of the $Ir_3^{4+}O_{12}$ trimer is responsible for the antiferromagnetic transition of Ba₄CeIr₃O₁₂. Specific heat measurements on Ba₄CeIr₃O₁₂ also show a clear anomaly against temperature at 10.5 K, indicating the occurrence of the long-range antiferromagnetic ordering at this temperature. Below 5 K, the magnetic susceptibility of Ba₄-CeIr₃O₁₂ increases with decreasing temperature, which is probably due to the existence of small amount of paramagnetic



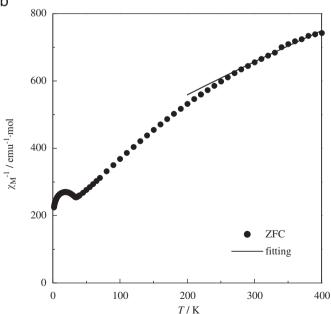


Fig. 7. (a) Temperature dependence of the magnetic susceptibilities for $Ba_4PrIr_3O_{12}$. The inset shows the magnetic susceptibilities at low temperatures. (b) The reciprocal magnetic susceptibility against temperature. The solid line is the Curie–Weiss fitting.

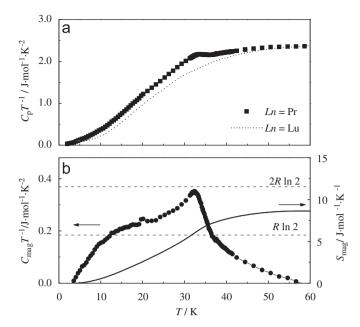


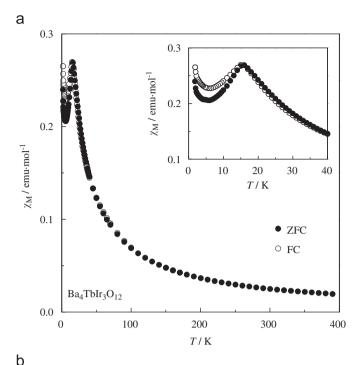
Fig. 8. Temperature dependence of (a) the specific heat divided by temperature (C_p/T) and (b) the magnetic specific heat divided by temperature $(C_{\rm mag}/T)$ and the magnetic entropy $(S_{\rm mag})$ for Ba₄PrIr₃O₁₂.

impurities such as Ba₃Celr₂O₉. The Ba₃Celr₂O₉ is paramagnetic down to 1.8 K [25].

3.2.2.2. $Ba_4PrIr_3O_{12}$. The results of the magnetic susceptibility measurements (Fig. 7(a)) show that $Ba_4PrIr_3O_{12}$ orders antiferromagnetically at 35 K. Its transition temperature is higher than that of $Ba_4CeIr_3O_{12}$. This reflects the fact that the Pr^{4+} ion contributes to the magnetic superexchange interaction between Ir^{4+} ions. This compound also shows the increase of magnetic susceptibility with decreasing temperature below 10 K. This behavior is ascribed to the paramagnetic impurities such as $Ba_3PrIr_2O_{9}$, which is paramagnetic down to 1.8 K [25].

In order to obtain detailed information about the low-temperature magnetic ordering, specific heat measurements were performed down to 1.8 K. Fig. 8(a) shows the temperature dependence of the specific heat (C_p) divided by temperature (C_p/T) for Ba₄PrIr₃O₁₂. A specific heat anomaly has been observed at 35 K, which corresponds to the results by magnetic susceptibility measurements. This result indicates that the long range antiferromagnetic ordering occur at this temperature. To evaluate the magnetic contribution to the specific heat (C_{mag}), we have to subtract the contribution of lattice specific heat (C_{lat}) from the total specific heat $(C_{mag} = C_p - C_{lat})$. The lattice specific heat was estimated by using the specific heat data of a diamagnetic compound Ba₄LuIr₃O₁₂ (the dotted line of Fig. 8(a)). Fig. 8(b) shows the temperature dependence of the magnetic specific heat divided by temperature (C_{mag}/T). The magnetic entropy change due to the magnetic ordering of Ba₄PrIr₃O₁₂ (S_{mag}) is calculated by the relation $S_{\text{mag}} = \int (C_{\text{mag}}/T)dT$. Its temperature dependence is also shown in Fig. 8(b). The total magnetic entropy change is obtained to be 8.7 J/mol K.

The corresponding ruthenium-containing compound $Ba_4PrRu_3O_{12}$ shows an antiferromagnetic transition at 2.4 K, and its specific heat measurements indicated that the antiferromagnetic ordering is only due to the ground Kramers doublet of Pr^{4+} ion [30]. The crystal structure of $Ba_4PrIr_3O_4$ is the same as that of $Ba_4PrRu_3O_4$. In a similar way, the ground Kramers doublet of Pr^{4+} ion in a low symmetric crystal field is responsible for the antiferromagnetic ordering of $Ba_4PrIr_3O_4$. Therefore, its contribution to the magnetic entropy change is estimated to be $R \ln 2$. The



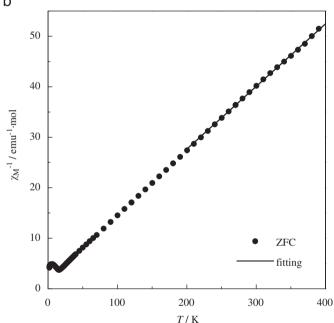


Fig. 9. (a) Temperature dependence of the magnetic susceptibilities for $Ba_4TbIr_3O_{12}$. The inset shows the magnetic susceptibilities at low temperatures. (b) The reciprocal magnetic susceptibility against temperature. The solid line is the Curie–Weiss fitting.

magnetic entropy change due to the antiferromagnetic ordering of the $Ir_3^{4+}O_{12}$ trimer (S=1/2) is also calculated to be $R\ln 2$. Although the magnetic entropy change obtained experimentally ($S_{\rm mag} = \sim 8.7 \, \text{J/mol K}$) is a little smaller than that calculated $2 \, R \ln 2$ ($= 11.52 \, \text{J/mol K}$), we can conclude that both the 4f electron of Pr^{4+} and the 5d electron of Ir^{4+} contribute to the antiferromagnetic ordering of $Ba_4PrIr_3O_4$.

 $3.2.2.3.\ Ba_4TbIr_3O_{12}$. Fig. 9(a) shows that Ba₄TbIr₃O₁₂ orders antiferromagnetically at 16 K. Below this temperature, the divergence between the ZFC and FC susceptibilities is observed, which indicates a small ferromagnetic moment. For compounds

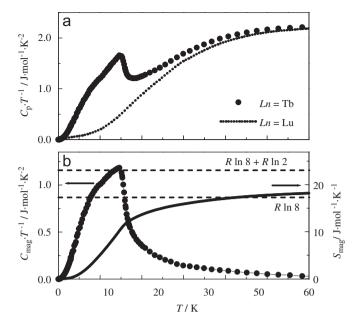


Fig. 10. Temperature dependence of (a) the specific heat divided by temperature (C_p/T) and (b) the magnetic specific heat divided by temperature $(C_{\rm mag}/T)$ and the magnetic entropy $(S_{\rm mag})$ for Ba₄Tblr₃O₁₂.

with low crystal symmetry such as monoclinic, a Dzyaloshinsky-Moriya (D-M) interaction can exist between the ordered magnetic moments, which results in the existence of a weak ferromagnetic moment associated with the antiferromagnetism. The increase of magnetic susceptibility with decreasing temperature below 5 K is probably due to the paramagnetic impurities such as Ba₃TbIr₂O₉ and Ba₂TbIrO₆ [25,42]. Fig. 10(a) shows the temperature dependence of the specific heat divided by temperature (C_p/T) for $Ba_4TbIr_3O_{12}$. A clear λ -type specific heat anomaly has been observed at 16 K, which indicates that the long range antiferromagnetic ordering occurs at this temperature. In the same way as the case for Ba₄PrIr₃O₁₂, the magnetic specific heat (C_{mag}) was evaluated by subtracting the specific heat of Ba₄LuIr₃O₁₂ (C_{lat}) from the total specific heat ($C_{mag} = C_p - C_{lat}$). Fig. 10(b) shows the temperature dependence of the magnetic specific heat divided temperature (C_{mag}/T). The magnetic entropy change for Ba₄TbIr₃O₁₂ (S_{mag}) is also shown in the same figure, and it is obtained to be \sim 18.5 J/mol K. Magnetic entropy change due to the magnetic ordering of Tb⁴⁺ (the electronic configuration of Tb⁴⁺: [Xe] f^7 ; the S = 7/2 ground state) is $R \ln(2S+1) =$ 17.3 J/mol K. The experimental value is larger than this value, which indicates the contribution of the $Ir_3^{4+}O_{12}$ trimer (S=1/2) to the magnetic ordering at 16 K. However, the experimental value is below the estimated total magnetic entropy change (23.1 J/mol K). The reason for this may be that the magnetic entropy is partially lost at higher temperatures because of the short range ordering. Similar smaller magnetic entropy change was reported for Ba₄TbRu₃O₁₂, i.e., the magnetic entropy change due to the magnetic ordering of Tb⁴⁺ was 12.5 J/mol K [30].

4. Summary

New quadruple perovskites $Ba_4LnIr_3O_{12}$ were prepared and they form the perovskite-type structure with 12 layers, in which

 Ir_3O_{12} trimers and LnO_6 octahedra are alternately linked by corner-sharing. For Ln= Ce, Pr, and Tb compounds, an antiferromagnetic transition was observed at low temperatures, while the other compounds are paramagnetic down to 1.8 K.

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